# Synthesis and Optical Properties of Y<sub>3-x</sub>Lu<sub>x</sub>Mg<sub>2</sub>AlSi<sub>2</sub>O<sub>12</sub>:Ce<sup>3+</sup>

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### Introduction

Fachhochschule

The most widely applied phosphor in white LEDs is cerium doped yttrium aluminum garnet (YAG:Ce). It shows strong absorption in the blue and broad emission band in the yellow spectral region. The position of the Ce<sup>3+</sup> emission band depends on crystal-field strength, covalency and Stokes shift. It is known that substitution of Y at dodecahedral sites by larger cations results in a red shift of emission, whereas smaller cations causes a blue shift. The opposite result is observed for octahedral and tetrahedral sites. The larger cation introduced into octahedral or tetrahedral site leads to a blue shift of emission and a smaller one a red shift. Another option for a blue or red shift of YAG:Ce emission band is substitution of yttrium by divalent or tetravalent cation, respectively. However, it is quite hard to find a large enough tetravalent cation for the voluminous dodecahedral sites. On the other hand, there are plenty of divalent cations suitable for substitution of yttrium. We show that substitution of 2Al<sup>3+</sup> by Mg<sup>2+</sup> - Si<sup>4+</sup> pair leads to a red shift of the Ce<sup>3+</sup> emission band. The present findings can be explained by the increased covalent character of the host lattice.





## Fig. 2. XRD patterns of Y<sub>3</sub>Mg<sub>2</sub>AlSi<sub>2</sub>O<sub>12</sub>:Ce<sup>3+</sup>

### Fig. 3. Reflection spectra of $Y_3Mg_2AISi_2O_{12}:Ce^{3+}$ samples

#### Wavelength (nm)

Fig. 4. Reflection spectra of  $Y_{3-x}Lu_{x}Mg_{2}AlSi_{2}O_{12}:2\%Ce^{3+}$  samples

#### Wavelength (nm)

Fig. 5. Excitation and emission spectra of Lu<sub>3</sub>Mg<sub>2</sub>AlSi<sub>2</sub>O<sub>12</sub>:Ce<sup>3+</sup> samples and their lumen equivalents



### **Results and Discussion**

Ce<sup>3+</sup>. Further increase of the dopant concentration

### Conclusions

All samples were synthesized by a sol-gel combustion method as depicted in **Fig. 1**. Sintering powders at 1400 C resulted in formation of garnet phase with solely minor impurities. A further increase of the annealing temperature resulted in reduction of impurity concentration (see Fig. 2). Phosphors possessed a yellow to orange body color indicating the absorption in the blue what is in line with the reflection spectra represented in **Fig. 3** and **Fig. 4**. As expected the absorption in the blue increased with higher  $Ce^{3+}$  concentration. emission of Excitation spectra and  $Lu_3Mg_2AISi_2O_{12}:Ce^{3+}$  samples are depicted in **Fig. 5**. The maximum excitation and emission intensity was achieved if phosphors were doped by 1%

led to a decrease of intensity and a red shift of emission maximum due to re-absorption of emitted photons by the activator. It turned out that emission spectra of Ce<sup>3+</sup> are a sensitive function of the Y/Lu ratio in the structure (see Fig. 6). Fig. 7 shows the thermal quenching of  $Y_3Mg_2AISi_2O_{12}$ :0.5%Ce<sup>3+</sup> sample. It reveals that emission intensity decreases if temperature is increased. It turned out that pending phosphors looses half of efficiency at 385 K (≈112 C). Color points of synthesized phosphors are shown in Fig. 8. It is obvious that the increase of Ce<sup>3+</sup> concentration leads to a redshift of the color point, what is in line with emission spectra. Finally, quantum efficiencies were calculated and are presented in **Fig. 9**. It turned out that QE gradually decreased if Ce<sup>3+</sup> concentration or Y<sup>3+</sup> content in the structure was increased.

In this work we demonstrated that the incorporation of Mg<sup>2+</sup> and Si<sup>4+</sup> ions onto the octahedral and tetrahedral sites, respectively, has led to red-shifted emission in comparison to the YAG:Ce phosphor. However, strong concentration and thermal quenching has been observed, which is a bottleneck to application yet.

